

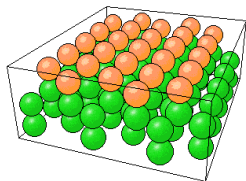
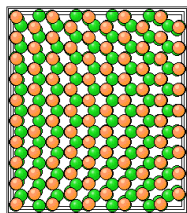
# Derivative-Free Optimization Methods for a Surface Structure Determination Problem

Juan C. Meza  
University of California, Merced

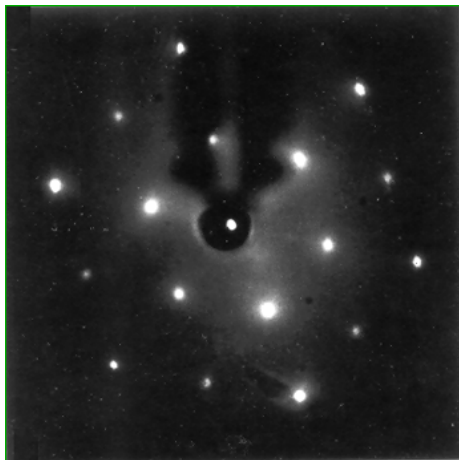
Applied Math Seminar  
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# Surface structure determination

- Surface structure determines many important properties of materials
- Possibility of electron diffraction first proposed by deBroglie (1924)
- Experimental data could not be quantitatively described by kinematic theory
- Over 40 years before electron diffraction became a tool in surface structure determination
- Necessitated the development of a theory of multiple scattering (late 1960s)

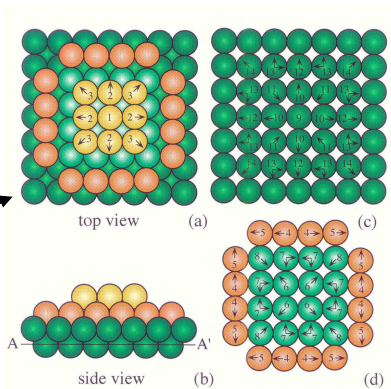
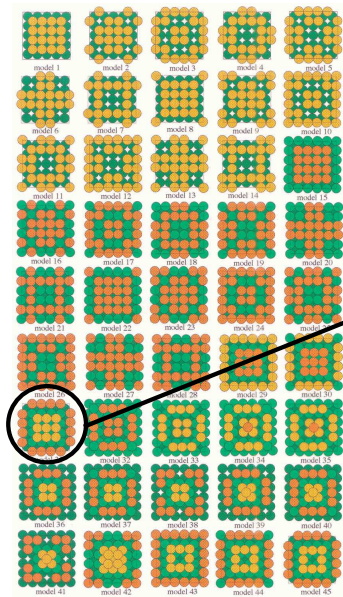


# Low-energy electron diffraction (LEED)



- Goal is to determine surface structure through low energy electron diffraction (LEED).
- Need to determine the coordinates and chemical identity of each atom.
- Ex: Li atoms on a Ni surface
- Low-energy electrons have high surface sensitivity, requiring a well-ordered surface

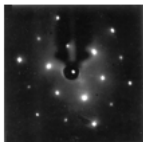
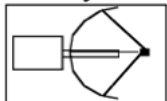
# What is the correct atomic configuration?



# Experimental setup

## Experiment

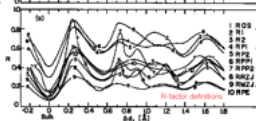
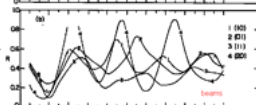
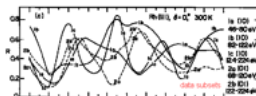
LEED system



I-V spectra



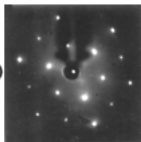
## R-Factors



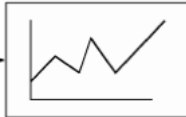
## Theory

(x,y,z) input parameters

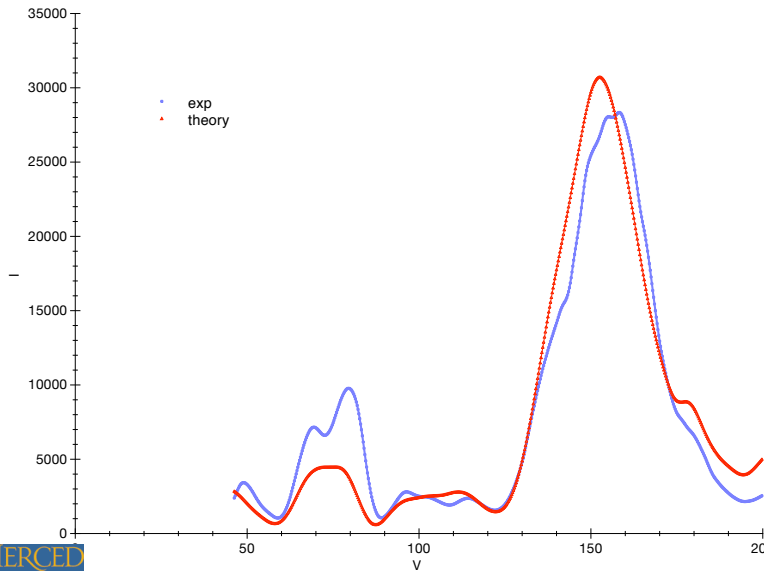
- 1) (-1.33, -0.08, 2.51)
- 2) (0.33, 0.00, 0.00)
- 3) (1.89, 1.22, 3.51)



I-V spectra



## Example IV curve



# Pendry Reliability-factor (1980)

- IV curves consist for the main part of a series of Lorentzian peaks ( $I$  is Intensity):

$$I \approx \sum \frac{a_j}{(E - E_i)^2 + V_{oi}^2}$$

- Pendry R-factor emphasizes positions of the peaks rather than the heights of the intensities
- Ideal agreement corresponds to  $R = 0$ ; uncorrelated spectra yields  $R = 1$ .

$$L(E) = I'/I$$

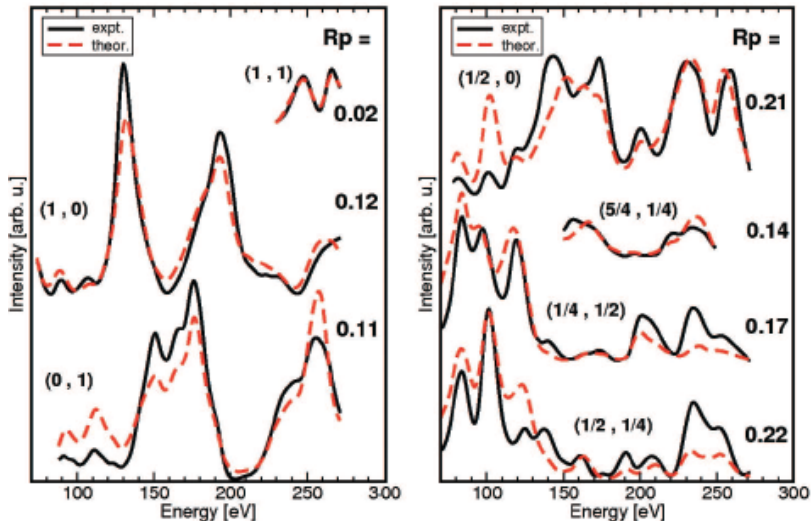
$$L \approx \sum_i \frac{-2(E - E_i)}{(E - E_i)^2 + V_{oi}^2}$$

$$Y(E) = L/[1 + (LV_{oi})^2]$$

$$R = \frac{\sum_{i,g} (Y_{th} - Y_{exp})^2}{\sum_{i,g} (Y_{th}^2 + Y_{exp}^2)}$$

where  $g$  denotes one beamline and  $E_i$  is the energy.

# R factors for various IV curves



Held, G., Low-energy electron diffraction crystallography of surfaces and interfaces. Bunsen-Magazin 12 (12), 2010. pp. 124-131.



# General Optimization Formulation

$$\begin{aligned} \min \quad & f(x), \quad x \in R^n \\ & h(x) = 0 \\ & g(x) \geq 0 \end{aligned}$$

For our problem:

- $f(x)$  = R-factor,  $x$  are coordinates of the atoms
- Several ways of computing the R-factor, we will use the Pendry R-factor.
- Sometimes referred to as parameter estimation or inverse problem

# Some standard assumptions

- Theoretical
  - Objective function is smooth
  - Derivatives are also "nice"
  - Constraints are linearly independent and smooth
- Practical
  - First and (sometimes) second derivatives are available
  - Objective and constraint functions are cheap to evaluate
  - Objective function has infinite (machine) precision

## Let's check our assumptions

- Function may be discontinuous
- Invalid (unphysical) structures lead to function being undefined in certain regions and returning **special values**
- No derivatives available; Hessians even harder to compute
- Objective and constraint function evaluations take up a substantial total of run time
- Function is a result of a computer simulation fitted to experimental data, i.e. less than infinite precision

And just for fun let's also add

- Combination of continuous and categorical variables
  - Atomic coordinates:  $x, y, z$
  - Chemical identity: Ni, Li

# Previous Work

- Early attempts used Hooke-Jeeves, nonlinear least squares, genetic algorithms<sup>1-2</sup>
- Effective, but expensive
  - Several 100,000s of function calls typically needed
  - Each function call can take several minutes on a workstation class computer
- We have also used pattern search methods – better, but still expensive

<sup>1</sup> Global Optimization in LEED Structure Determination Using Genetic Algorithms, R. Dill and M.A. Van Hove, Surf. Sci. 355, L393-8 (1996).

<sup>2</sup> G. S. Stone, MS dissertation, Computer Science Dept., San Francisco State University, 1998.

# What is a Pattern Search Method?

- Particular instance of Direct Search method - methods that “in their heart” do not use gradient information, e.g. Nelder-Mead simplex (M. Wright, 1996).
- Can also be classified as a **Generating Set Search** method
- Main operation is function comparisons
- Useful whenever the derivative is not available or too expensive to compute, for example so called **simulation-based** optimization problems
- Unlike other DFO methods, these methods are strictly monotonic (vs. GA or SA)

# General observations

- Use multiple search directions in such a way as to ensure at least one is a descent direction
- Makes the methods ideal for parallel computation
- Can use simple or sufficient decrease
- Never uses gradient in practice, but theory does require gradient be Lipschitz continuous or continuously differentiable

# General Algorithm

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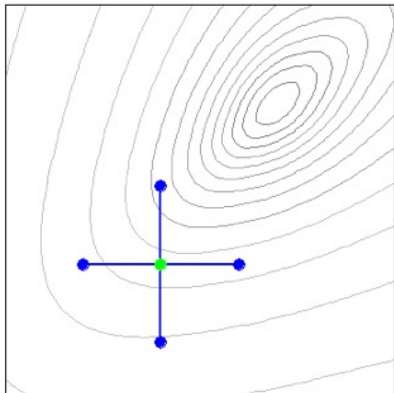
Initialization: Given values  $\Delta_0, x_0, M_0, P_0$ ,

```
1   For  $k = 0, \dots, \text{maxit}$  do
2       Search: Evaluate  $f$  on a finite subset of trial points on a mesh  $M_k$ .
3       Poll: Evaluate  $f$  on the frame  $P_k$ .
4       If (successful)
5            $x_{k+1} = x_k + \Delta_k d_k$ 
6       Update  $\Delta_k$ 
7   End
```

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- Search phase can include user heuristics or surrogate functions
- Poll phase is more rigid, but needed to ensure convergence, i.e. sufficient descent directions.

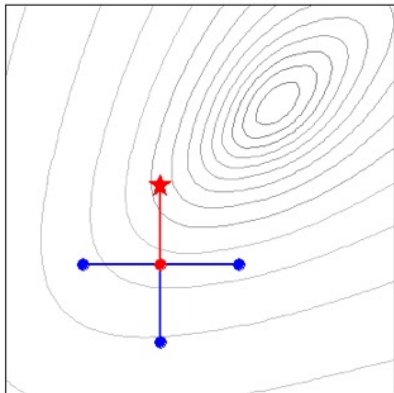
# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



Special thanks to Tammy Kolda for this slide

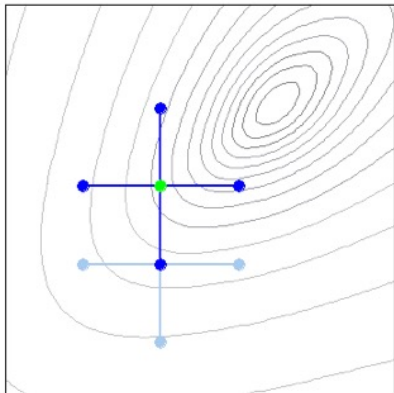


# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



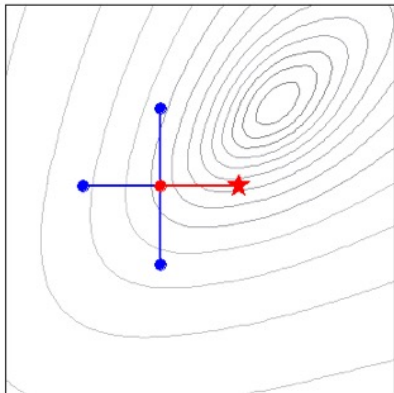
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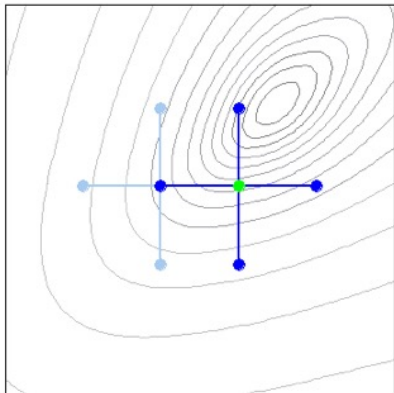
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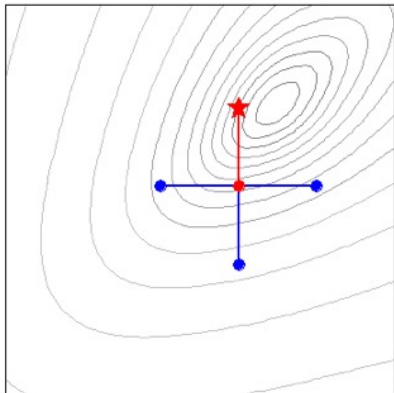
Special thanks to Tammy Kolda for this slide

# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



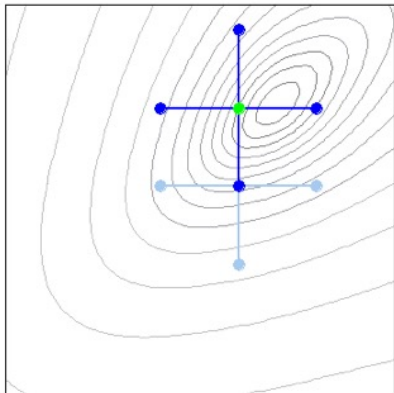
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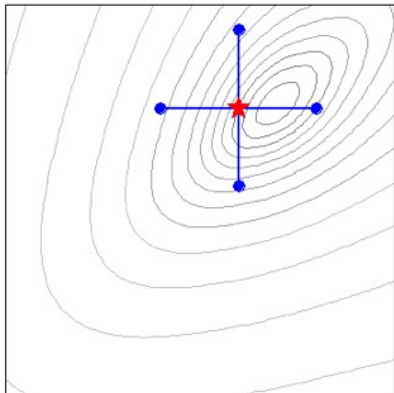
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# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



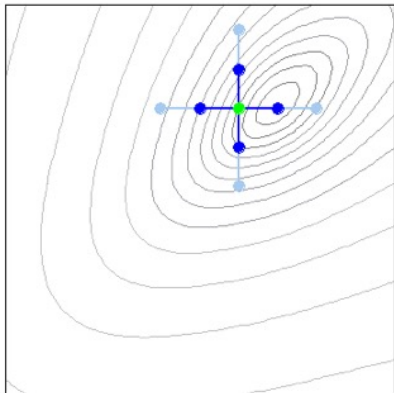
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# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



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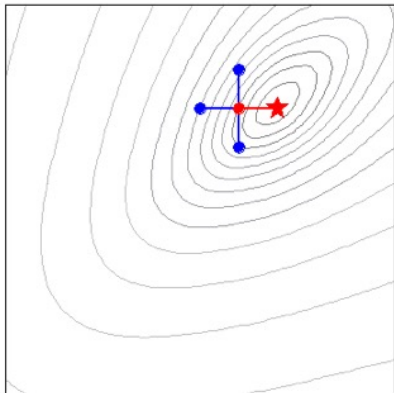
# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



Special thanks to Tammy Kolda for this slide



# Pattern Search $D_k = \{\pm e_1, \pm e_2\}$



# Global Convergence Properties, Part I

Generating Set Search (GSS) globalization strategies hinge on one key concept:

For all *unsuccessful* iterations

$$\lim_{k \rightarrow \infty} \|\Delta_k\| = 0$$

Various ways to produce such a sequence including

- sufficient decrease
- rational lattices
- moving grids

General assumptions mild including either  $f$  is bounded below or level sets are bounded.

## Global Convergence Properties, Part II

If  $f(x)$  is suitably smooth (continuously differentiable) can show that

- For unsuccessful iterations,  $\|\nabla f(x_k)\|$  is bounded as a function of the step length  $\Delta_k$
- And since we can ensure  $\liminf \|\Delta_k\| = 0$

### Global Convergence

$$\liminf_{k \rightarrow \infty} \|\nabla f(x_k)\| = 0$$

N.B. We can prove stronger results under stricter conditions.

# General Algorithm

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Initialization: Given values  $\Delta_0, x_0, M_0, P_0$ ,

```
1   For  $k = 0, \dots, \text{maxit}$  do
2       Search: Evaluate  $f$  on a finite subset of trial points on a mesh  $M_k$ .
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4       If (successful)
5            $x_{k+1} = x_k + \Delta_k d_k$ 
6       Update  $\Delta_k$ 
7   End
```

---

- Search phase can include user heuristics or surrogate functions
- Poll phase is more rigid, but needed to ensure convergence, i.e. sufficient descent directions.

# Taking advantage of LEED

- LEED

- Multiple scattering model
- I-V spectra computed repeatedly until best-fit structure is found
- Computation time is proportional to the number of parameters
- Most expensive of all methods

- KLEED

- Assumes that electrons are only scattered once by the surface atoms.
- Agrees well with experimental data as long as the incident angle is close to perpendicular
- Surface unit cell size and symmetry can be determined, but not the exact positions
- Compared to multiple scattering, KLEED I-V spectra can be computed order of magnitude faster

Idea: Use KLEED as a simplified physics surrogate

# Building a better surrogate

- Define  $\phi_A(x) = \phi_S + \phi_I$ , where
  - $\phi_A$  = Additive surrogate
  - $\phi_S$  = Simplified physics surrogate, e.g. KLEED
  - $\phi_I$  = Interpolatory surrogate
- Interpolatory surrogate designed to capture difference between simplified physics and “true” function values

## Search:

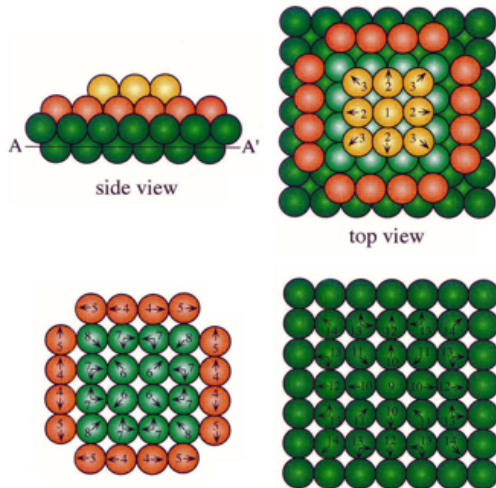
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```
1  if (first search step) {  
2      Initialize  $\phi_I$  with Latin Hypercube sample.  
3  }  
4      recalibrate  $\phi_I$  with DACE model of  $\phi_S - f$ .  
5  }  
6  Construct  $\phi_A = \phi_S + \phi_I$   
7  Solve min  $\phi_A$ 
```

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# Test problem

Ni(100)-(5x5)-Li



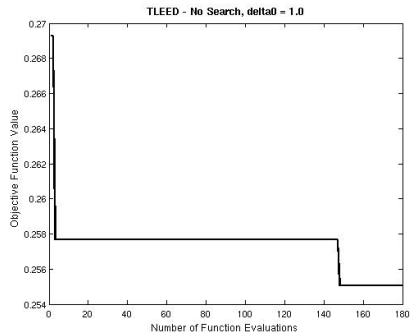
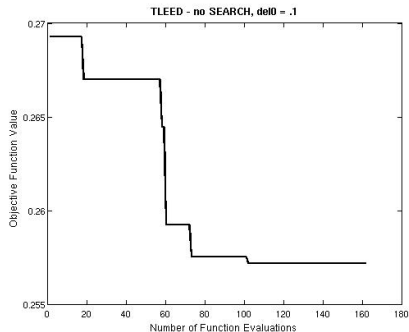
- Model 31 from set of model problem using three layers
  - 14 atoms
  - 14 categorical variables
  - 42 continuous variables
- Additional constraint added so that positions of atoms are constrained to lie within a box
- Used NOMADm (Abramson, Audet, Dennis, Le Digabel, Tribes)

# Test cases

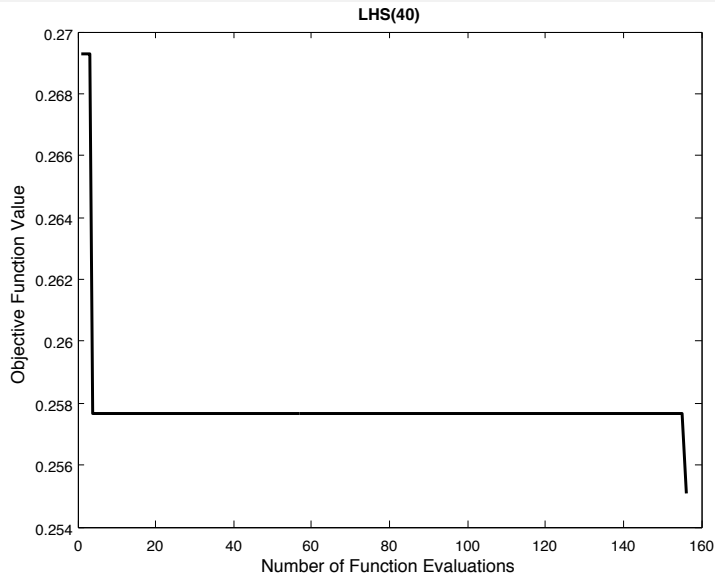
- Start with best known feasible point
- Continuous variables only
- Three different approaches
  - No search step
  - LHS search
  - Simplified physics surrogate (DACE model)



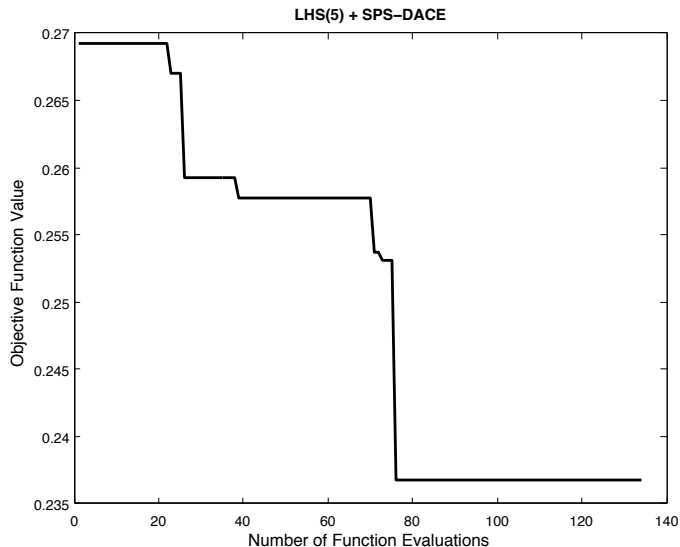
# Optimization of continuous variables using no search phase



## Optimization using LHS with 40 points



# Optimization using additive surrogate, $\Delta_0 = 0.1$



## Summary of numerical results

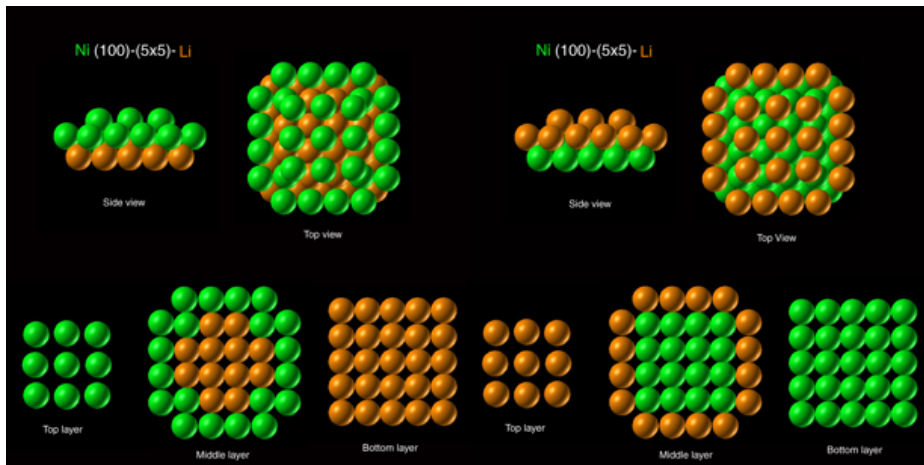
Method	LHS	$f(x^*)$	fevals
No search	0	0.2551	180
LHS	40	0.2551	160
SPS+DACE	15	0.2543	180
SPS+DACE	5	0.2354	135

# Summary

- Simulation-Based Optimization is an important and rapidly growing area of research
- Many standard assumptions do not apply for this class of problems – great opportunity for new ideas and research
- Presented one example that used a combination of adapting old methods and taking advantage of structure
  - Total number of function evaluations decreased by about 20% per model
  - Room for improvement and new ideas, e.g. alternate interpolatory surrogates
  - Another direction is solving the full mixed variable problem

Questions?

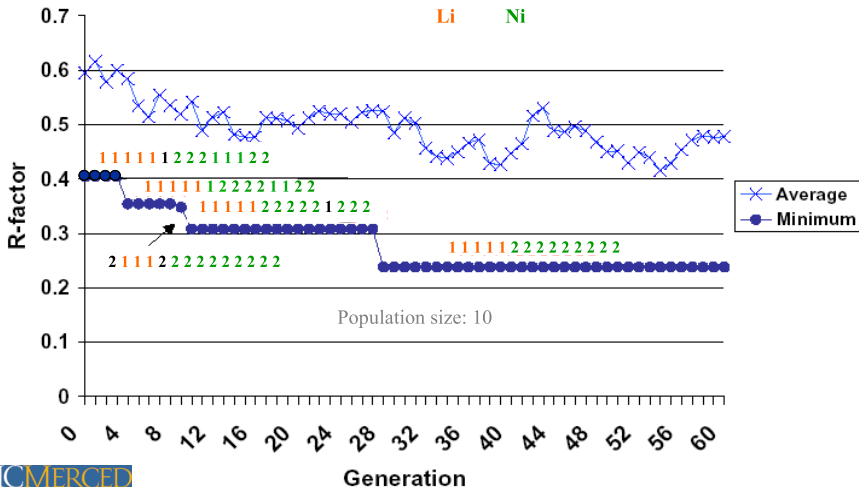
# Future Work – Chemical Identity Search



# GA with Chemical Identity

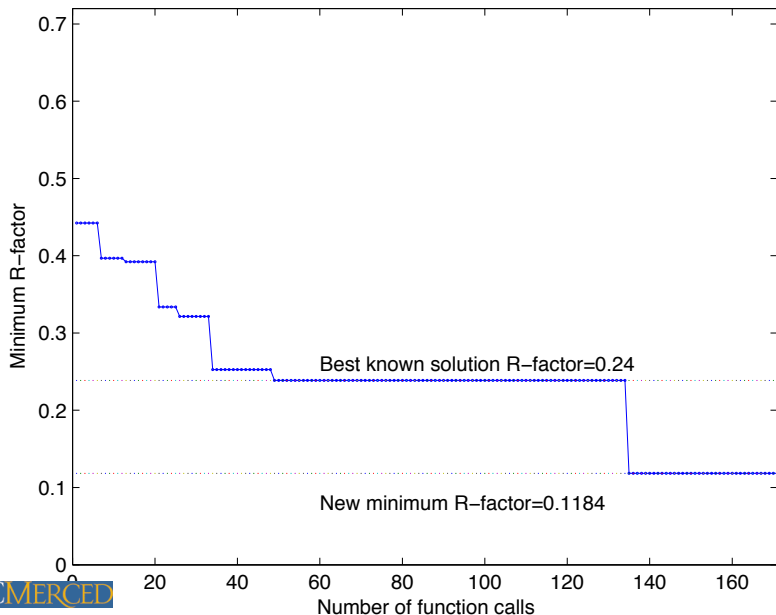
## LEED Chemical Identity Search: Ni(001)-(5x5)-Li

best known solution: 11111222222222  
Li Ni





# Pattern Search: Chemical Identity + Cont. Vars



## Pendry Reliability-factor (1980)

$$R = \sum_{i,g} (Y_{gth} - Y_{gexp})^2 / \sum_{i,g} (Y_{gth}^2 + Y_{gexp}^2)$$

$$Y(E) = L / (1 + LV_{oi}^2)$$

$$L(E) = I' / I$$

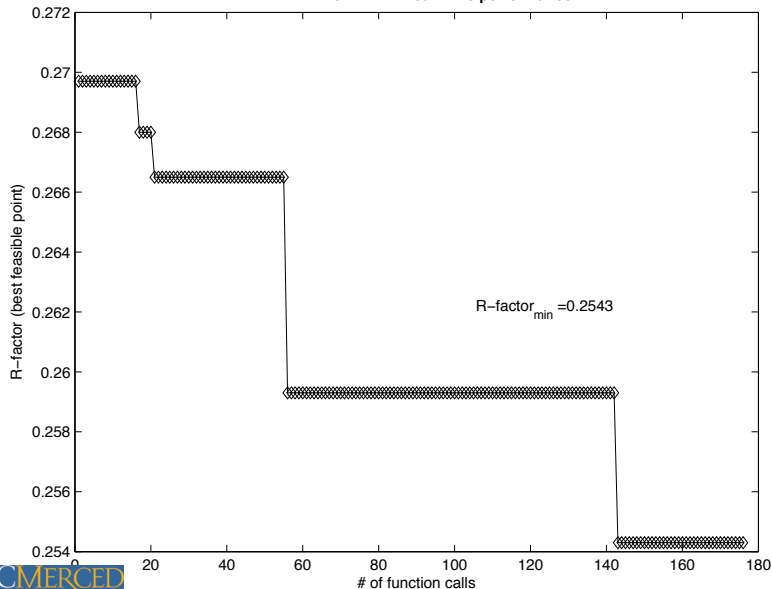
$$L \approx \sum_j \frac{-2(E - E_j)}{(E - E_j)^2 + V_{oi}^2}$$

# Pendry R-factor

- LEED curves consist for the main part of a series of Lorentzian peaks:  
$$I \approx \sum a_j / (E - E_j)^2 + V_{oi}^2$$
- Their widths are dictated by the imaginary part of the electron self-energy (optical potential):  $\Delta E = 2|V_{oi}|$
- Pendry R-factor emphasizes positions of the maximum and minimum rather than the heights of the intensities
- Ideal agreement corresponds to  $R = 0$ ; uncorrelated spectra yields  $R = 1$ .

# Optimization using additive surrogate, $\Delta_0 = 1.0$

TLEED-NOMADm: Real-time performance



# Kinematic LEED

- KLEED assumes that electrons are only scattered once by the surface atoms.
- Agrees well with experimental data as long as the incident angle is close to perpendicular.
- Surface unit cell size and symmetry can be determined, but not the exact positions.
- Compared to multiple scattering which takes several minutes to compute, I-V spectra from KLEED can be computed in a few seconds.

Idea: Use KLEED as a simplified physics surrogate

# Pendry Reliability-factor (1980)

- IV curves consist for the main part of a series of Lorentzian peaks ( $I$  is Intensity):

$$I \approx \sum \frac{a_j}{(E - E_j)^2 + V_{oi}^2}$$

- Pendry R-factor emphasizes positions of the peaks rather than the heights of the intensities
- Ideal agreement corresponds to  $R = 0$ ; uncorrelated spectra yields  $R = 1$ .

$$L(E) = I'/I$$

$$L \approx \sum_j \frac{-2(E - E_j)}{(E - E_j)^2 + V_{oi}^2}$$

$$Y(E) = L/[1 + (LV_{oi})^2]$$

$$R = \frac{\sum_{i,g} (Y_{gth} - Y_{gexp})^2}{\sum_{i,g} (Y_{gth}^2 + Y_{gexp}^2)}$$